



## Crystalline phase engineering on cocatalysts: A promising approach to enhancement on photocatalytic conversion of carbon dioxide to fuels

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### ABSTRACT

Photocatalytic conversion of CO<sub>2</sub> to value-added fuels, a promising route to address the depletion of fossil fuels and concomitant global warming, can be improved by cocatalysts through promoting the electron-hole separation and offering catalytic active sites for the surface reactions. However, the enhancement of the photocatalytic performance is greatly determined by the surface and interface structures of cocatalysts. Herein, for the first time, we demonstrate that the photocatalytic activity in the CO<sub>2</sub> reduction can be optimized through crystalline phase design of cocatalysts. In this work, Ru nanocrystals in face-centered cubic (fcc) and hexagonal close-packed (hcp) phases are *in-situ* grown on C<sub>3</sub>N<sub>4</sub> nanosheets to form different C<sub>3</sub>N<sub>4</sub>-Ru hybrid structures, respectively. It was found that the hcp Ru achieved higher average CO and CH<sub>4</sub> production rate but lower H<sub>2</sub> production rate in comparison with fcc Ru. As revealed by the experimental characterizations combined with theory simulations, the phase-dependent photocatalytic performance is resulted from the different surface reaction behaviors on the fcc and hcp Ru cocatalysts. The adsorption energy of CO<sub>2</sub> molecules on the dominated (1011) face of hcp Ru is higher than that on the dominated (111) face of fcc Ru. As a result, the stronger interaction between CO<sub>2</sub> molecules and the surface of hcp Ru contributes to the enhanced photocatalytic activity and selectivity of C<sub>3</sub>N<sub>4</sub>-hcp Ru in reduction of CO<sub>2</sub>. This work highlights the importance in the crystalline phase engineering in cocatalyst for enhanced CO<sub>2</sub> photoreduction.

### 1. Introduction

To address the increasing energy and environmental related concerns, production of chemical fuels utilizing renewable solar energy has received extensive attention [1–3]. Particularly, photocatalytic conversion of CO<sub>2</sub> to value-added chemicals, such as methane (CO<sub>2</sub> + 8H<sup>+</sup> + 8e<sup>−</sup> → CH<sub>4</sub> + 2H<sub>2</sub>O) and carbon oxide (CO<sub>2</sub> + 2H<sup>+</sup> + 2e<sup>−</sup> → CO + H<sub>2</sub>O) is of particular interest because of its capability of mimicking natural photosynthesis through a green and renewable route [4–11]. Nowadays, a variety of semiconductor based photocatalysts such as TiO<sub>2</sub> [12,13], C<sub>3</sub>N<sub>4</sub> [14], CdS [15], CeO<sub>2</sub> [16], Zn<sub>2</sub>GeO<sub>4</sub> [17], ZnGa<sub>2</sub>O<sub>4</sub> [18] and BiWO<sub>6</sub> [19] have been studied for CO<sub>2</sub> reduction reaction. However, the major challenges so far in the technology of photocatalytic CO<sub>2</sub> reduction are serious electron-hole

recombination in bulk semiconductor photocatalysts and low adsorption and activation abilities of stable CO<sub>2</sub> molecules on semiconductor surface [20]. In view of this restriction, metal and alloy cocatalysts, such as Pd [21], Pt [22,23], Ag [24], Rh [25], AuCu [26], and PtCu [27], were loaded on the semiconductor surface for enhanced photocatalytic CO<sub>2</sub> conversion efficiency. On one hand, metal cocatalysts trap the photogenerated electrons from semiconductor in promoting the electron-hole separation. On the other hand, the surface of cocatalysts offers highly reactive sites in promoting the adsorption and activation of CO<sub>2</sub> molecules [28]. Graphitic-phase carbon nitride (g-C<sub>3</sub>N<sub>4</sub>), a typical visible light-excitable n-type polymeric semiconductor was widely used as the cocatalyst supporter in the photocatalytic CO<sub>2</sub> reduction reaction [29–31]. The large surface area of C<sub>3</sub>N<sub>4</sub> nanosheets facilitates the loading of cocatalyst, while the low thickness shortens the electron

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transfer distance to the cocatalysts and reduces the possibility of electrons loss. In addition, the conjugated aromatic structure of  $\text{C}_3\text{N}_4$  offers manifold coupling configurations with cocatalysts, ensuring a strong connection between them for high-efficient interfacial electron transfer [14].

In spite of the efforts, the photocatalytic activity and selectivity of semiconductor supported metal cocatalyst hybrid structures in  $\text{CO}_2$  reduction reaction is still far from practical applications. For instance, resulted from the high  $\text{H}_2\text{O}$  activation ability on the surface of cocatalysts, the reduction of  $\text{H}_2\text{O}$  to  $\text{H}_2$  ( $2\text{H}_2^+ + 2\text{e}^- \rightarrow \text{H}_2$ ) is a preferential or competitive process, which lowers the product yields and selectivity [32]. Considering the two key roles of cocatalysts in the photocatalytic process, surface and interface design of cocatalysts is a promising route to further enhance the photocatalytic activity and selectivity in  $\text{CO}_2$  reduction [33–35]. Generally, it is believed that the design of cocatalysts with high electron trapping ability from semiconductor and abundant surface active sites for the adsorption and activation of  $\text{CO}_2$  molecules is vital to realize high-performance photocatalysis. On the other hand, the adsorption and activation abilities of  $\text{H}_2\text{O}$  molecules on the surface of designed cocatalysts should be reduced to decrease the side reaction.

In the surface and interface design of cocatalysts in photocatalysis, crystallographic structure, namely the arrangement of atoms in the nanocrystals, is an important parameter. On one hand, crystallographic structure plays a critical role in influencing the electronic band structure of cocatalyst. Thus the electron trapping ability of cocatalyst from light-harvesting semiconductor is greatly dependent on its crystallographic phase [36,37]. On the other hand, different crystallographic structures result in different atomic arrangements on the surface of cocatalyst. Thus reactive sites in different types and densities are exposed on the surface of cocatalysts for catalytic reaction [38,39]. Herein, for the first time, we demonstrate that the photocatalytic performance in  $\text{CO}_2$  reduction reaction can be optimized through crystalline phase engineering on metal cocatalysts. In this paper, ruthenium (Ru) nanocrystals in face-centered cubic (fcc) and hexagonal close-packed (hcp) phases are *in-situ* grown on  $\text{C}_3\text{N}_4$  nanosheets to form  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru semiconductor-cocatalyst hybrid structures, respectively. Based on the experimental results and theoretical simulations, it is found that though fcc Ru and hcp Ru possess the comparable electron trapping ability from  $\text{C}_3\text{N}_4$ , the surface of hcp Ru cocatalyst exhibits superior surface adsorption and activation abilities of  $\text{CO}_2$  molecules in comparison with that of fcc Ru. Furthermore, a lower surface catalytic ability in reduction of  $\text{H}_2\text{O}$  to  $\text{H}_2$  is realized by hcp Ru. As a result, the as-obtained  $\text{C}_3\text{N}_4$ -hcp Ru exhibits significantly superior photocatalytic activity and selectivity in conversion of  $\text{CO}_2$  to fuels as compared with  $\text{C}_3\text{N}_4$ -fcc Ru.

## 2. Experimental

### 2.1. Chemicals

$\text{Ru}(\text{acac})_3$  (Aladdin, R118502),  $\text{RuCl}_3 \cdot \text{nH}_2\text{O}$  (Aladdin, R109234), triethylene glycol (TEG, Aladdin, T110409) and ethylene glycol (EG, Sigma-Aldrich, 324558) were used in our synthesis. All other chemicals were of analytical grade and purchased from Sinopharm Chemical Reagent Co., Ltd. without further purification. The water used in all experiments was de-ionized.

### 2.2. Synthesis of $\text{C}_3\text{N}_4$ nanosheets

Bulk g- $\text{C}_3\text{N}_4$  was synthesized through a modified method according to previous reports [27,40]. In a typical synthesis, 20 g of urea was dissolved in 10 mL water. The obtained solution was heated at 550 °C for 3 h (0.5 °C/min ramp rate; natural cooling). Then the resultant yellow agglomerates were milled into g- $\text{C}_3\text{N}_4$  powder in a mortar. Then the as-synthesized g- $\text{C}_3\text{N}_4$  powder was further dispersed in TEG and EG

to form the corresponding suspension of  $\text{C}_3\text{N}_4$  nanosheets (15 mg/mL) with probe sonication for 1 h, respectively.

### 2.3. Synthesis of $\text{C}_3\text{N}_4$ -Ru samples

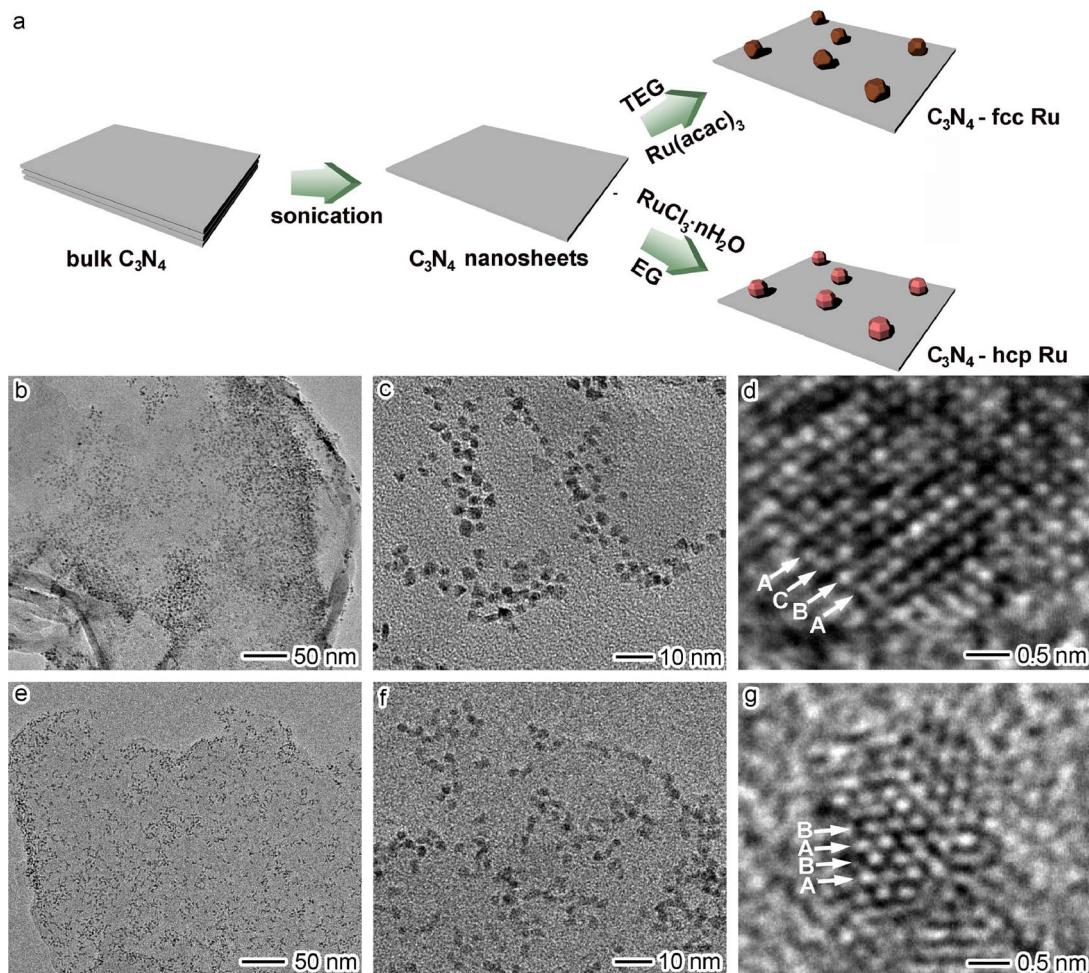
In a typical synthesis of  $\text{C}_3\text{N}_4$ -fcc Ru, 0.21 mmol of  $\text{Ru}(\text{acac})_3$  and 0.5 mmol of Poly(vinyl pyrrolidone) (PVP, K30, monomer units) were dissolved in 10 mL of TEG suspension of  $\text{C}_3\text{N}_4$  at room temperature. The dispersion was then heated to 200 °C and maintained at this temperature for 3 h. After completion of the reaction, the resultant product was separated by centrifugation, washed with water three times, and dried at 45 °C in vacuum. In the synthesis of  $\text{C}_3\text{N}_4$ -hcp Ru, 0.21 mmol of  $\text{RuCl}_3 \cdot \text{nH}_2\text{O}$  and 0.5 mmol of PVP were dissolved in 10 mL of EG suspension of  $\text{C}_3\text{N}_4$  at room temperature. The dispersion was then heated to 200 °C and maintained at this temperature for 3 h. The as-obtained product was separated by centrifugation, washed with water three times, and dried at 45 °C in vacuum.

### 2.4. Sample characterizations

X-ray powder diffraction (XRD) patterns were recorded by using a Philips X'Pert Pro Super X-ray diffractometer with  $\text{Cu-K}\alpha$  radiation ( $\lambda = 1.54178 \text{ \AA}$ ). X-ray photoelectron spectra (XPS) were collected on an ESCALab 250 X-ray photoelectron spectrometer, using non-monochromatized Al-K $\alpha$  X-ray as the excitation source. Transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) were taken on a JEOL JEM-2100F field-emission high-resolution transmission electron microscope operated at 200 kV. The concentrations of metal elements were measured as follows: the samples were dissolved with a mixture of  $\text{HNO}_3$  and  $\text{HCl}$  (1:3, volume ratio), which was then diluted with 1%  $\text{HNO}_3$ . The concentrations of metal ions were then measured through a Thermo Scientific PlasmaQuad 3 inductively-coupled plasma mass spectrometry (ICP-MS). The loading amounts of Ru on the  $\text{C}_3\text{N}_4$  nanosheets were determined by sample weighing prior to the dissolution of Ru for the ICP-MS measurements. UV-vis-NIR diffuse reflectance data were measured in the spectral region of 200–800 nm with a Shimadzu SolidSpec-3700 spectrophotometer. Photoluminescence (PL) spectra were carried out on a HITACHI F-7000 Spectrofluorometer with the excitation wavelength of 390 nm. Fourier transform infrared (FTIR) measurements were performed on a Nicolet 8700 FTIR spectrometer in a KBr pellet, scanning from 4000 to 500  $\text{cm}^{-1}$ . The electron spin resonance (ESR) spectra were characterized with a JES-FA200 electron spin resonance spectrometer by mixing the sample with 5,5-dimethyl-1-pyrroline-N-oxide (DMPO) (100  $\mu\text{L}$ , 50 mM) in water under Xe lamp irradiation.

### 2.5. Photoelectrochemical measurements

3.0-mg as-synthesized products were dispersed in a mixture of 10- $\mu\text{L}$  nafion and 10- $\mu\text{L}$  ethanol, which were then uniformly spin-dropped onto a 1 cm × 1 cm indium tin oxide (ITO)-coated glass. Subsequently, the ITO-coated glass was heated at 80 °C for 1 h in a vacuum oven. The photocurrents were measured on a CHI 760E electrochemical station (Shanghai Chenhua, China) under irradiation of a 300-W Xe lamp (Solaredge 700, China). Visible light ( $780 \text{ nm} > \lambda > 420 \text{ nm}$ ) was used as the illumination source, which was realized by using both a 420-nm cutoff filter (long-wave-pass) and a 780-nm cutoff filter (short-wave-pass). The power density of visible light was measured to be 100  $\text{mW cm}^{-2}$ . Standard three-electrode setup was used with the ITO coated glass as photoelectrode, a Pt foil as counter electrode, and an Ag/AgCl electrode as reference electrode. Three electrodes were inserted in a quartz cell filled with 0.5-M  $\text{Na}_2\text{SO}_4$  electrolyte. The photo-response of as-prepared photoelectrodes (i.e.,  $I-t$ ) was operated by measuring the photocurrent densities under chopped light irradiation (light on/off cycles: 60 s) at a bias potential of 0.2 V vs. Ag/AgCl for 400 s. The electrochemical impedance spectroscopy (EIS) was carried



**Fig. 1.** (a) Schematic illustrating the synthesis of  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru; (b–d) TEM (b and c) and HRTEM (d) images of  $\text{C}_3\text{N}_4$ -fcc Ru; (e–g) TEM (e and f) and HRTEM (g) images of  $\text{C}_3\text{N}_4$ -hcp Ru.

out in the frequency range of  $10^{-1}$ – $10^5$  Hz with an AC voltage amplitude of 10 mV at a bias potential of 0.2 V vs. Ag/AgCl.

## 2.6. Photocatalytic $\text{CO}_2$ reduction measurements

$\text{CO}_2$  photocatalytic reduction reaction was typically conducted in a 100 mL reactor purchased from Beijing Perfect Light Company (China). To remove possible trace organic contaminants, all the samples were treated at 200 °C for 5 h in air. In a typical measurement, 15 mg of photocatalyst was dispersed on the flat glass plate on the bottom of the reactor. And 2 mL  $\text{H}_2\text{O}$  was added and surrounded the plate. The reactor loaded with catalysts was first purged with high purity  $\text{CO}_2$  for 30 min. Then the light-irradiation experiment was performed by using a 300 W Xe lamp with visible light as the illumination source. The power density of visible light ( $780 \text{ nm} > \lambda > 420 \text{ nm}$ ) was measured to be 100 mW  $\text{cm}^{-2}$ . The photocatalytic reaction was typically performed for 4 h. During the irradiation, 1 mL gas was sampled from the glass chamber at a given time interval (1 h). The amount of products evolved were determined using gas chromatography (GC-2014, Shimadzu) with Ar as carrier gas. The  $\text{H}_2$  and  $\text{O}_2$  were determined using a thermal conductivity detector (TCD). The  $\text{CH}_4$  were measured by flame ionization detector (FID) and the CO was converted to  $\text{CH}_4$  by a methanation reactor and then analyzed by the FID. During the stability test, the photocatalysts were collected after each run and then reused for photoreduction of  $\text{CO}_2$ .

## 2.7. Computational methods

The theoretical works were calculated by the Cambridge Sequential Total Energy Package (CASTEP) in Materials Studio Package [41,42]. The ultrasoft pseudopotential with a frozen-core approximation was adopted for the electron-ion interaction [43]. The generalized gradient approximation (GGA) [44] was used for the electronic exchange and correlation. The Perdew-Burke-Ernzerhof functional was used in this work [45]. The Kohn-Sham one-electron states were expanded in a plane wave basis. The cutoff energy and smearing value were set as 400 and 0.1 eV, respectively. Brillouin zone integration was approximated by a sum over special  $k$ -points using the Monkhorst-Pack method [46]. The energy convergence criterion was set to  $2.0 \times 10^{-6}$  eV/atom; while the maximum force and displacement tolerances of structure optimization were set to 0.05 eV/ $\text{\AA}$  and  $5 \times 10^{-3}$   $\text{\AA}$ , respectively [47].

The calculated lattice constants of hcp Ru are 2.720 and 4.284  $\text{\AA}$ , which are well consistent with the experimental value, 2.706 and 4.282  $\text{\AA}$ , respectively. The fcc lattice constant is 3.807  $\text{\AA}$ , which is also agreement with the experimental one, 3.830  $\text{\AA}$ . According to the simulation of fcc and hcp Ru nanoparticles [39], the possible exposed surfaces of the fcc Ru(111), (100), (221), (211), (311), (110), (210), (310), (321) and (320), as well as the hcp Ru(0001), (110), (1012), (111), (201), (1010), (1011) and (2130) were used for the investigation of  $\text{CO}_2$  adsorption behaviors. The vacuum space was set as 12  $\text{\AA}$  between the slabs to minimize their interaction. Free  $\text{CO}_2$  was optimized in a cubic lattice with the length of 15  $\text{\AA}$  [48].

$\text{CO}_2$  adsorption behaviors at different sites on substrate were

systematically studied, and the corresponding adsorption energy was defined as:

$$E_{\text{ads}} = E_{\text{CO}_2/\text{substrate}} - (E_{\text{CO}_2} + E_{\text{substrate}}) \quad (1)$$

where  $E_{\text{CO}_2/\text{substrate}}$  is the total energy of the substrate with  $\text{CO}_2$ ;  $E_{\text{CO}_2}$  and  $E_{\text{substrate}}$  are the energies of free  $\text{CO}_2$  and substrate, respectively [27]. Therefore, a negative value means exothermic adsorption, while a positive one means endothermic adsorption. The more negative the adsorption energy, the stronger the adsorption [49].

### 3. Results and discussion

#### 3.1. Sample synthesis and characterization

In the typical synthesis of the samples, bulk  $\text{C}_3\text{N}_4$  was firstly prepared with urea as precursor (Fig. S1), which was then exfoliated into  $\text{C}_3\text{N}_4$  nanosheets via a sonication treatment (Fig. 1a) [27,40]. As shown in Fig. S2, the exfoliated  $\text{C}_3\text{N}_4$  nanosheets are in wrinkles with micrometer-sized surface and nanoscale thickness. In the design of the hybrid photocatalysts,  $\text{C}_3\text{N}_4$  as light harvesting semiconductor can be excited by visible light and produce photogenerated electron-hole pairs, in which electrons can transfer to the surface of metal cocatalysts for  $\text{CO}_2$  reduction reaction through the formation of Schottky junction [28].

In the second step, Ru nanocrystals in fcc phase were then *in situ* grown on the flat surface of  $\text{C}_3\text{N}_4$  nanosheets through reducing Ru (acac)<sub>3</sub> with TEG acting as both the reducing agent and solvent, while hcp Ru nanocrystals were *in situ* grown on  $\text{C}_3\text{N}_4$  nanosheets with  $\text{RuCl}_3\text{nH}_2\text{O}$  as Ru source as well as EG as reducing agent and solvent (Fig. 1a) [50]. As shown in the TEM images (Fig. 1b and e), Ru nanocrystals were uniformly distributed over the flat surface of  $\text{C}_3\text{N}_4$  nanosheets in both the samples without any agglomerations. The Ru nanocrystals are in narrow size distribution and the average sizes of them are determined to be 3.7 and 3.0 nm in  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru samples, respectively (Figs. 1c and f, S3). HRTEM image in the  $\text{C}_3\text{N}_4$ -fcc Ru clearly reveals the fcc ABCABCABC... stacking sequence in the monocrystalline Ru nanocrystals, while a hcp ABABAB... stacking sequence is observed in the HRTEM image of Ru nanocrystals in the  $\text{C}_3\text{N}_4$ -hcp Ru sample (Fig. 1d and g), indicating the different crystalline phases of Ru cocatalysts between the two samples.

The crystal structures of the as-obtained  $\text{C}_3\text{N}_4$  based samples were further revealed by X-ray powder diffraction (XRD) pattern (Fig. 2a). For bare  $\text{C}_3\text{N}_4$ , weak (100) peak at 13.1° represents the in-planar repeating tri-s-triazine unit, while the strong (002) peak at 27.4° is the characteristic interlayer stacking reflection of the conjugated aromatic segments [51]. As for the XRD pattern of  $\text{C}_3\text{N}_4$ -fcc Ru, additional peaks are in good agreement with the fcc Ru (JCPDS No. 88-2333). In contrast, additional peaks in the XRD pattern of  $\text{C}_3\text{N}_4$ -hcp Ru sample are indexed to the hcp Ru (JCPDS No. 06-0663). The XRD results confirm the combination of  $\text{C}_3\text{N}_4$  with Ru in different phases in the two samples. It should be noted that there is a slight shift of the XRD peaks of  $\text{C}_3\text{N}_4$ -hcp Ru to smaller angles compared with the JCPDS data with the possible reason of some lattice distortions in the hcp Ru nanoparticles. The chemical structures of the samples were further indicated by Fourier transform infrared (FTIR) spectra (Fig. 2b). It can be clearly seen that the all the characteristic peaks, including the N–H stretching (3000–3600  $\text{cm}^{-1}$ ), stretching vibrations of CN heterocycles (1000–1800  $\text{cm}^{-1}$ ) and breathing mode of the triazine units (810  $\text{cm}^{-1}$ ), are observed in the spectra of bare  $\text{C}_3\text{N}_4$  [52]. In comparison with the FTIR spectrum of bare  $\text{C}_3\text{N}_4$ , no additional peak was shown in the spectra of  $\text{C}_3\text{N}_4$ -Ru samples, suggesting that there is no obvious change of  $\text{C}_3\text{N}_4$  during the further growth of Ru nanocrystals. The compositions of the as-obtained  $\text{C}_3\text{N}_4$ -Ru samples were further determined by X-ray photoelectron spectra (XPS). As shown in Fig. 2c, the survey XPS spectrum of  $\text{C}_3\text{N}_4$ -fcc Ru indicates the existence of C, N and Ru elements. The C1s high-resolution spectrum can be deconvoluted

into two peaks at 284.6 and 288.0 eV, which are assigned to the graphitic carbon (C–C) and  $\text{sp}^2$ -bonded aromatic structure (N–C=N), respectively (Fig. 2d) [53]. In the high-resolution XPS spectrum of N1s (Fig. 2e), the binding energies at 398.4, 400.2, and 404.1 eV correspond to the  $\text{sp}^2$ -hybridized aromatic N (C=N–C), tertiary N bonded to carbon atoms [N–(C)<sub>3</sub>] and  $\pi$ -excitations, respectively [54]. As for the high-resolution Ru3p spectrum (Fig. 2f), the doublet centered at 461.0 and 483.2 eV are assigned to the Ru3p<sub>3/2</sub> and Ru3p<sub>1/2</sub> of Ru (0), revealing the metallic Ru in the  $\text{C}_3\text{N}_4$ -fcc Ru sample [55]. The XPS spectra of  $\text{C}_3\text{N}_4$ -hcp Ru are much similar to those of  $\text{C}_3\text{N}_4$ -fcc Ru, confirming the same chemical composition of the two  $\text{C}_3\text{N}_4$ -Ru samples (Fig. S4). In facilitating the comparison of following photocatalytic performance, the loading amounts of Ru on the  $\text{C}_3\text{N}_4$  were kept the same in the two samples, which was confirmed by the inductively coupled plasma mass spectrometry (ICP-MS) measurement (Table S1).

#### 3.2. Charge kinetics and photocatalytic performance analysis

Considering the photocatalytic activity is greatly determined by the generation and transfer of photo-induced charge carriers, the charge kinetics behaviors of the  $\text{C}_3\text{N}_4$  based samples were investigated prior to the photocatalytic performance analysis. The light absorption capabilities of the samples were first revealed by UV-vis-NIR diffuse reflectance spectra. As shown in Fig. 3a, bare  $\text{C}_3\text{N}_4$  exhibits a light absorption in visible region with an adsorption edge at *ca.* 460 nm. In comparison bare  $\text{C}_3\text{N}_4$ , the  $\text{C}_3\text{N}_4$ -Ru samples exhibit additional light absorption at longer wavelengths, which is resulted from the excitation of bound electrons to high energy levels through interband transitions of non-plasmonic Ru [56,57]. More importantly, the two  $\text{C}_3\text{N}_4$ -Ru samples exhibit similar light absorption properties in visible region, revealing their comparable abilities in generating electron-hole pairs under visible light irradiation. Then the electron-hole separation abilities of the samples were further analyzed by the photocurrent response under visible light irradiation. As shown in Fig. 3b, the photocurrent of  $\text{C}_3\text{N}_4$  is much lower than those of  $\text{C}_3\text{N}_4$ -Ru samples, indicating that Ru nanocrystals as cocatalysts can trap the photogenerated electrons from  $\text{C}_3\text{N}_4$  in promoting the electron-hole separation. The smooth charge transfer between the electron donor of  $\text{C}_3\text{N}_4$  and electron acceptor of Ru elucidates the strong metal-semiconductor interaction between them. More importantly,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru display approximate photocurrent response, suggesting the similar electron trapping abilities of Ru cocatalysts in different phases. This statement is further confirmed by the EIS (Fig. 3c), in which  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru show approximative arc radii, revealing the similar resistance owing to the equivalent interfacial electron transfer ability from  $\text{C}_3\text{N}_4$  to Ru in different phases. Furthermore, the information of electron-hole separation can also be provided by PL emission spectra. Under the excitation wavelength of 390 nm,  $\text{C}_3\text{N}_4$  exhibits a PL emission with the maximum at 460 nm, which is quenched in almost the same degree by fcc and hcp Ru cocatalysts (Fig. 3d). In view of the PL of semiconductor is induced by the radiative electron-hole recombination, the similar PL quench degree reflects that the fcc and hcp Ru cocatalysts have approximate electron trapping abilities in suppressing the electron-hole recombination in  $\text{C}_3\text{N}_4$  [58].

Have confirmed the charge kinetics of the  $\text{C}_3\text{N}_4$  based samples, their photocatalytic performance in the reduction of  $\text{CO}_2$  with  $\text{H}_2\text{O}$  was further investigated. The samples were excited under visible light irradiation in the mixed vapor of  $\text{CO}_2$  and  $\text{H}_2\text{O}$ , and the amounts of products were determined by gas chromatography. As shown in Fig. 4a, bare  $\text{C}_3\text{N}_4$  produces  $\text{H}_2$ , CO and  $\text{CH}_4$ , indicating that the photo-generated electrons participate in both the conversion of  $\text{CO}_2$  to CO and  $\text{CH}_4$ , and the reduction of  $\text{H}_2\text{O}$  to  $\text{H}_2$ . According to the required electrons to generate reduction products using the following equation:

$$\text{Selectivity } (\%) = [8\nu(\text{CH}_4) + 2\nu(\text{CO})]/[2\nu(\text{CO}) + 8\nu(\text{CH}_4) + 2\nu(\text{H}_2)] \times 100\% \quad (2)$$

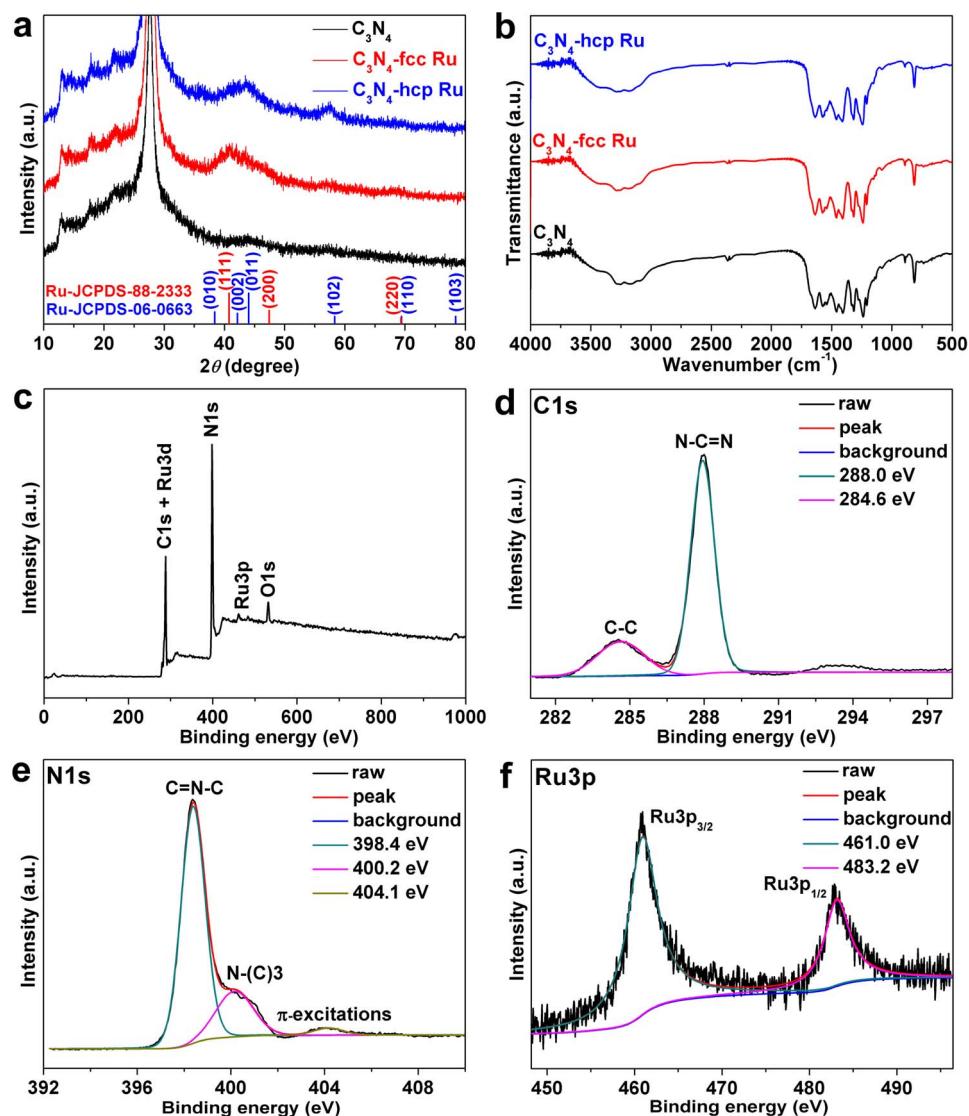


Fig. 2. (a) XRD patterns and (b) FTIR spectra of  $\text{C}_3\text{N}_4$ ,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru; (c–f) XPS spectra of  $\text{C}_3\text{N}_4$ -fcc Ru: (c) survey spectrum; (d) C1s; (e) N1s and (f) Ru3p high-resolution spectra.

the selectivity of  $\text{C}_3\text{N}_4$  in  $\text{CO}_2$  reduction is evaluated to be 38.5%. When the Ru nanocrystals are loaded on the light-harvesting  $\text{C}_3\text{N}_4$ , the average production rates of  $\text{H}_2$ ,  $\text{CO}$  and  $\text{CH}_4$  significantly increase, confirming the cocatalyst role of Ru in the photocatalytic process. More importantly,  $4.78 \mu\text{mol g}_{\text{cat}}^{-1} \text{h}^{-1}$  of average  $\text{CO}$  production rate and  $0.78 \mu\text{mol g}_{\text{cat}}^{-1} \text{h}^{-1}$  of average  $\text{CH}_4$  production rate are achieved by  $\text{C}_3\text{N}_4$ -hcp Ru, 1.6 and 2.3 times higher than those of  $\text{C}_3\text{N}_4$ -fcc Ru, respectively. Furthermore, it is interesting that the average  $\text{H}_2$  evolution rate of  $\text{C}_3\text{N}_4$ -hcp Ru is lower than that of  $\text{C}_3\text{N}_4$ -fcc Ru. As a result, the photocatalytic selectivity in  $\text{CO}_2$  reduction increases from 46.5% of  $\text{C}_3\text{N}_4$ -fcc Ru to 71.8% of  $\text{C}_3\text{N}_4$ -hcp Ru, revealing that more electrons generated in  $\text{C}_3\text{N}_4$ -hcp Ru participate in the photocatalytic  $\text{CO}_2$  reduction reaction. In addition to the high photocatalytic activity and selectivity, the  $\text{C}_3\text{N}_4$ -hcp Ru also exhibits excellent stability in the  $\text{CO}_2$  reduction reaction. There are only slight declines in the average production rate of  $\text{CO}$  and  $\text{CH}_4$  during the four cyclic processes, which may be resulted from the loss of photocatalysts during the collection of them after each run and reused for the next cycle as well as the fall off of minimal Ru nanocrystals from  $\text{C}_3\text{N}_4$  nanosheets during the cycles. Moreover, the photocatalytic selectivity in  $\text{CO}_2$  reduction maintains in the range of 70%–72.2%. From the TEM image of the sample after the cyclic process (Fig. S5), it can be clearly seen that most of Ru

nanocrystals are still firmly attached to the flat surface of  $\text{C}_3\text{N}_4$  nanosheets without any aggregation. Only a trace amount of Ru nanocrystals are detached from the  $\text{C}_3\text{N}_4$  nanosheets. Also in the XRD patterns and XPS spectra of  $\text{C}_3\text{N}_4$ -hcp Ru before and after the cycles, it can be seen that there is no obvious change in the spectra after the cycles except the slight decrease in the intensities of Ru peaks (Fig. S6). In spite of this, it was found that the photocatalysts can be repeated for more than 15 times (60 h). The high photocatalytic activity of  $\text{C}_3\text{N}_4$ -hcp Ru is also confirmed by the superior  $\text{CO}$  and  $\text{CH}_4$  production rate and higher selectivity in  $\text{CO}_2$  reduction in comparison with previous reported  $\text{C}_3\text{N}_4$  supported fcc metal cocatalyst structures, further showing the importance of crystalline phase of cocatalysts in the photocatalytic reaction (Table S2) [14,29,59–61]. It should be noted that the formation of  $\text{O}_2$  over the  $\text{C}_3\text{N}_4$  based samples was also confirmed by us, indicating that the photogenerated holes left in the  $\text{C}_3\text{N}_4$  also participate in the  $\text{H}_2\text{O}$  oxidation reaction ( $2\text{H}_2\text{O} + 4\text{h}^+ \rightarrow \text{O}_2 + 4\text{H}^+$ ). The generation of holes in the reaction is also confirmed by ESR spectra (Fig. S7), in which ESR signals with intensity ratio of 1:2:2:1 are clearly displayed after mixing the  $\text{C}_3\text{N}_4$ -hcp Ru with DMPO in water under visible light irradiation, suggesting that the holes participate in the production of hydroxyl radicals ( $\text{OH}$ ) [62,63]. Given the equivalent abilities of the two  $\text{C}_3\text{N}_4$ -Ru samples in generating and transferring

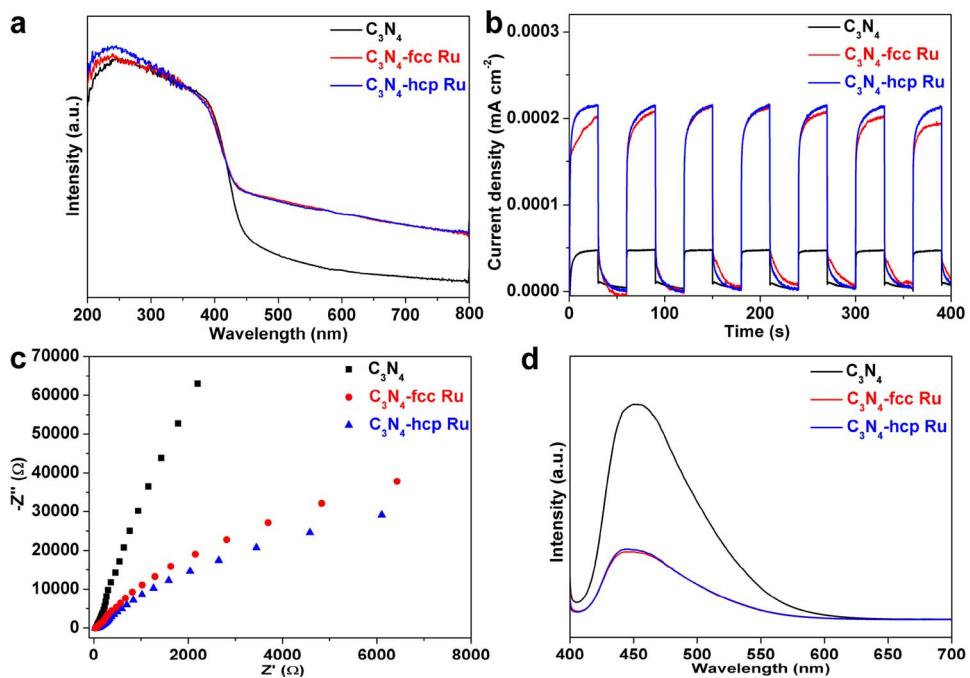


Fig. 3. (a) UV-vis-NIR diffuse reflectance spectra of  $\text{C}_3\text{N}_4$ ,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru; (b and c) photocurrent vs. time (I-t) curves (b) and EIS Nyquist plots (c) of  $\text{C}_3\text{N}_4$ ,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru at 0.2 V vs. Ag/AgCl under visible light ( $780 \text{ nm} > \lambda > 420 \text{ nm}$ ) irradiation; (d) PL spectra of  $\text{C}_3\text{N}_4$ ,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru excited at 390 nm.

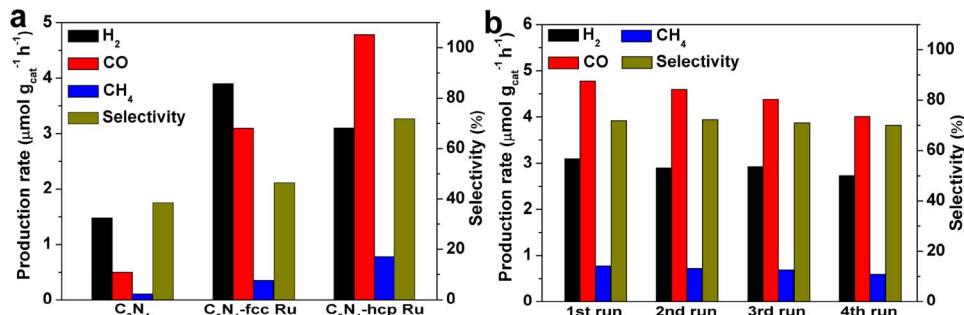


Fig. 4. (a) Average  $\text{H}_2$ ,  $\text{CO}$  and  $\text{CH}_4$  evolution rates of  $\text{C}_3\text{N}_4$ ,  $\text{C}_3\text{N}_4$ -fcc Ru and  $\text{C}_3\text{N}_4$ -hcp Ru photocatalysts as well as their selectivities in  $\text{CO}_2$  reduction under visible light ( $420 < \lambda < 780 \text{ nm}$ ) irradiation; (b) average  $\text{H}_2$ ,  $\text{CO}$  and  $\text{CH}_4$  evolution rates of  $\text{C}_3\text{N}_4$ -hcp Ru photocatalyst as well as its selectivity in  $\text{CO}_2$  reduction in four successive cycles under visible light ( $420 < \lambda < 780 \text{ nm}$ ) irradiation.

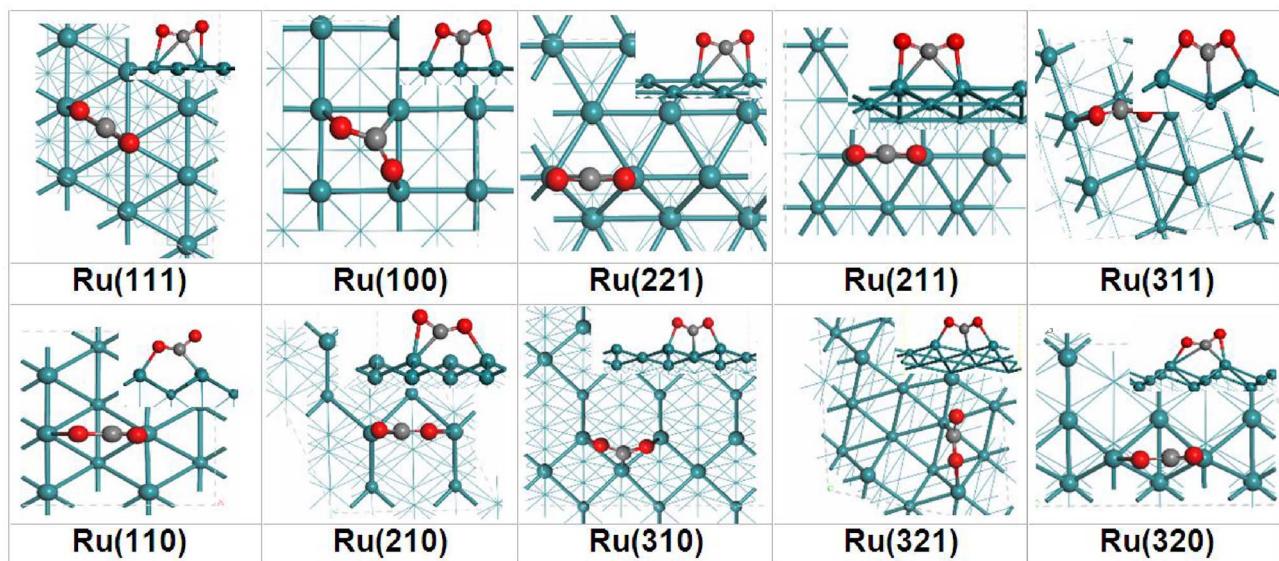


Fig. 5. Most stable configurations of  $\text{CO}_2$  on fcc Ru surfaces (blue-green, dark and red ones for Ru, C and O atoms, respectively). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

**Table 1**  
Parameters of CO<sub>2</sub> on the fcc Ru surfaces.

Surface	Coordination number	Work function/eV	E <sub>ads</sub> /eV	d <sub>C-O</sub> /Å	Mulliken charge
111	9	4.63	−0.12	1.252	1.262
100	8	4.63	−0.66	1.257	1.258
221	7	4.13	−0.84	1.257	1.260
211	7	4.07	−0.75	1.259	1.264
311	7	4.03	−1.24	1.293	1.297
110	7	3.96	−0.62	1.255	1.280
210	6	4.10	−1.01	1.268	1.276
310	6	4.02	−1.38	1.298	1.321
321	6	3.94	−1.27	1.294	1.298
320	6	3.84	−1.15	1.260	1.281

photo-induced electrons, the different photocatalytic performance may be resulted from the different reaction behaviors on the surface of Ru cocatalysts in different phases, which gives a good explanation of higher CO and CH<sub>4</sub> average production rates but lower H<sub>2</sub> production rate of C<sub>3</sub>N<sub>4</sub>-hcp Ru as compared with those of C<sub>3</sub>N<sub>4</sub>-fcc Ru.

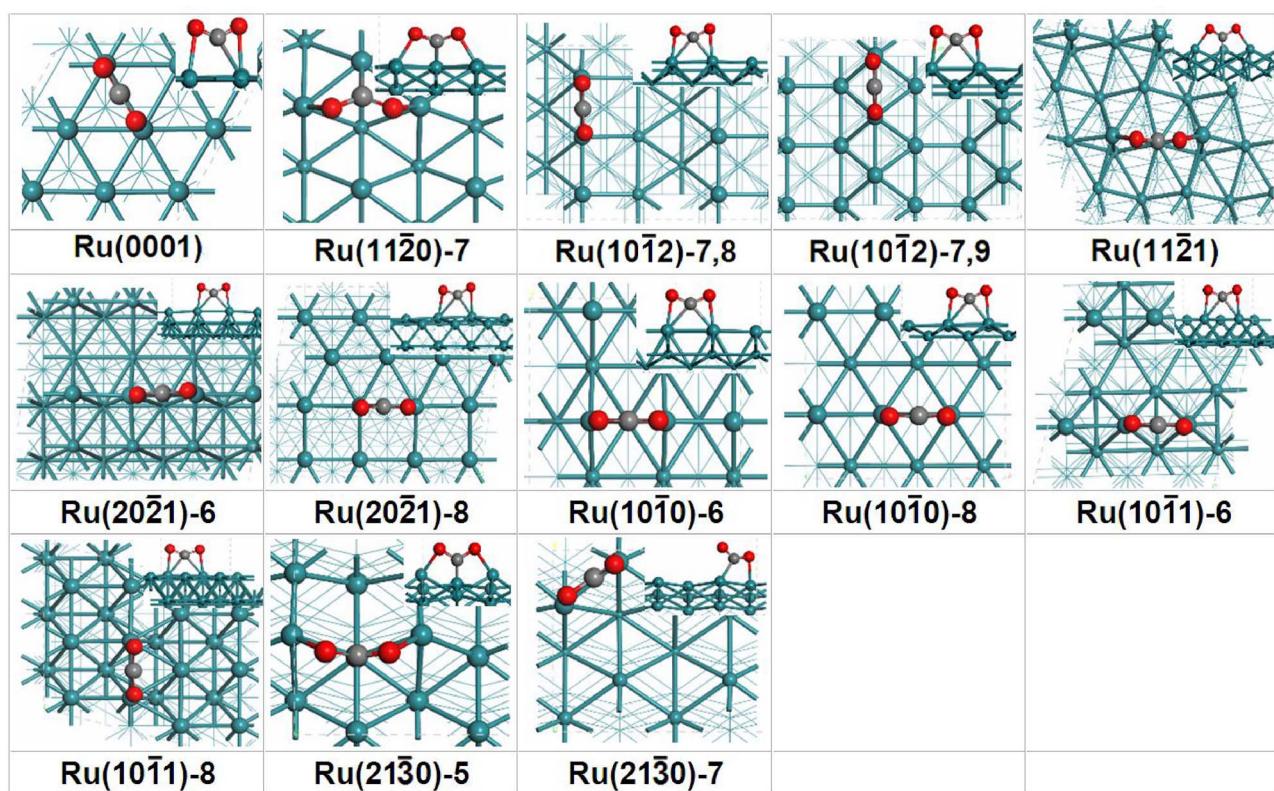
### 3.3. Photocatalytic mechanism analysis

To further analyze the possible mechanism of the phase-dependent photocatalytic performance in CO<sub>2</sub> reduction reaction, theoretical simulation was carried out by density functional theory (DFT) calculation. The adsorption behaviors of CO<sub>2</sub> on the surfaces of the fcc and hcp Ru nanoparticles were systematically investigated. The most stable configurations of CO<sub>2</sub> on the fcc surfaces were shown in Fig. 5 and the corresponding parameters were listed in Table 1, while the configurations on the hcp surfaces and the corresponding parameters were shown in Fig. 6 and Table 2, respectively. It can be found that the interactions between the surfaces of Ru cocatalysts with CO<sub>2</sub> molecules are highly

**Table 2**  
Parameters of CO<sub>2</sub> on the hcp Ru surfaces.

Surface	Coordination number	Work function/eV	E <sub>ads</sub> /eV	d <sub>C-O</sub> /Å	Mulliken charge
0001	9	4.33	−0.12	1.251	1.256
11̄20	7	4.09	−1.30	1.294	1.296
10̄12	7(8)	4.08	−0.93	1.261	1.268
	7(9)	4.15	−0.70	1.261	1.261
11̄21	6	3.95	−1.40	1.292	1.295
20̄21	6	3.93	−1.13	1.262	1.266
	8	4.05	−0.33	1.249	1.264
10̄10	6	3.69	−0.98	1.267	1.271
	8	4.38	−0.58	1.303	1.330
10̄11	6	3.66	−1.05	1.267	1.269
	8	4.38	−0.39	1.256	1.258
21̄30	5	3.93	−1.41	1.298	1.299
	7	4.03	−1.00	1.244	1.285

related with the coordination number and the atom arrangement of surface Ru atoms. For both the fcc and hcp Ru, it can be found that the interaction energies between CO<sub>2</sub> and Ru surfaces increase with the reduction of the coordination number [64]. With the high coordination number of 9, fcc Ru(111) and hcp Ru(0001) surfaces exhibit the same CO<sub>2</sub> adsorption energy (−0.12 eV). When the coordination number decreases to 8, the adsorption energy of CO<sub>2</sub> is −0.66 eV on fcc Ru (100) and −0.39 eV on hcp Ru(10̄11). The values further decrease to −1.30 to −0.62 eV on the surface of 7-coordinated Ru and −1.40 to −0.98 eV on the face of 6-coordinated Ru. Furthermore, the work function is also dependent on the atomic arrangement of Ru. Generally, lower coordination number of Ru contributes to a smaller work function. Though both the adsorption energy and work function are affected by the facet of Ru, the differences between the work functions are much smaller in comparison with those between the CO<sub>2</sub> adsorption energy.



**Fig. 6.** Most stable configurations of CO<sub>2</sub> on hcp Ru surfaces. Number after the surface index indicates the coordination number of surface Ru atom (blue-green, dark and red ones for Ru, C and O atoms, respectively). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

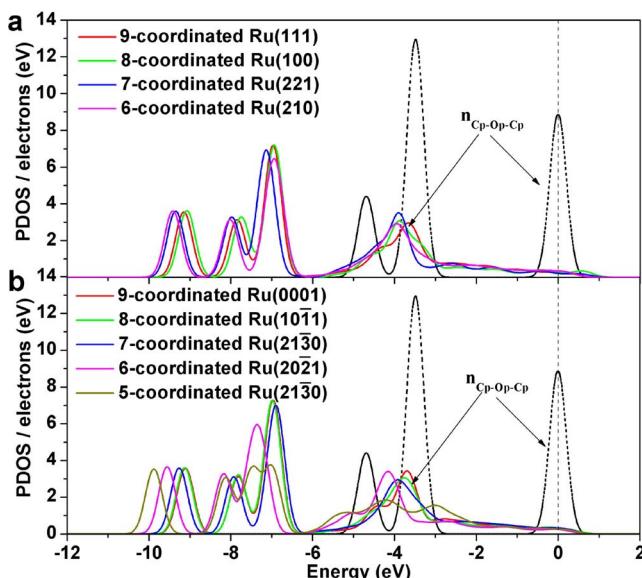


Fig. 7. PDOS of  $\text{CO}_2$  on the typical (a) fcc and (b) hcp Ru surfaces with different surface Ru coordination numbers. Short dash curve is for free  $\text{CO}_2$  molecule, straight line is for the Fermi level.

This conclusion is further supported by the partial density of states of  $\text{CO}_2$  adsorbed on Ru surfaces. As shown in Fig. 7, the HOMO and LUMO orbitals of free  $\text{CO}_2$  are the  $n_{\text{Cp-Op-Cp}}$  and  $\pi^*_{\text{Cp-Op-Cp}}$ , respectively. With the adsorption of  $\text{CO}_2$  molecules on the surface of Ru, its electronic peak shifts towards lower energy due to the interactions between  $\text{CO}_2$  and Ru surfaces. The peak area of HOMO orbital of  $\text{CO}_2$  decreases compared with that of free molecule, and the LUMO orbital of  $\text{CO}_2$  shifts downward to the Fermi energy level. These results indicate that  $\text{CO}_2$  loses electrons to Ru surface and the electrons on Ru surfaces feedback to  $\text{CO}_2$  simultaneously. Furthermore, it can be seen that the peak shifts increase with the decrease of Ru coordination number. This indicates the interactions between  $\text{CO}_2$  and Ru surfaces increase with the reduction of coordination number of Ru, which is in consistent with the results reported in the literature [65,66].

According to previous reported literature [39], based on Wulff construction from DFT calculated surface energies, fcc Ru is octahedron-like populated mainly by close-packed (111) facets, while hcp Ru is a dihedral-like shape dominated with (0001) and (10-11) surface (Fig. S8). Though the adsorption energies of  $\text{CO}_2$  on fcc Ru(111) and hcp Ru (0001) are almost the same ( $-0.12$  eV), the adsorption energy of  $\text{CO}_2$  on the hcp Ru(10-11) ( $-0.39$  eV) is much higher than that on fcc Ru (111) ( $-0.12$  eV). As a result, stronger interaction occurs between  $\text{CO}_2$  molecules and hcp Ru(10-11), which is further supported by the higher negative Mulliken charges of adsorbed  $\text{CO}_2$ . The Mulliken charges are  $-0.50\text{ e}^-$  on fcc Ru(111) and  $-0.53\text{ e}^-$  on hcp Ru(10-11), respectively (Tables 1 and 2). Furthermore, there are more edges and corners with low coordination atoms in the hcp Ru nanocrystals as compared with those in the fcc ones [39]. Considering the lower coordination number of Ru results in higher  $\text{CO}_2$  adsorption energy, the higher densities of edges and corners in hcp Ru are potential active sites for  $\text{CO}_2$  conversion. Therefore, the higher surface adsorption ability of  $\text{CO}_2$  as well as the higher density of active sites in the edges and corners contributes to the higher photocatalytic activity of the hcp Ru in reduction of  $\text{CO}_2$ , which is well consistent with the experimental results. It should be noted that the work function of fcc Ru(111) is  $4.63$  eV, much similar to those of hcp Ru(0001) ( $4.33$  eV) and Ru(10-11) ( $4.38$  eV), suggesting the approximate electron trapping ability of Ru cocatalysts from  $\text{C}_3\text{N}_4$  with lower work function ( $4.31$  eV) despite the different phases [14]. This result is in good agreement with the nearly identical probability of interfacial electron transfer from  $\text{C}_3\text{N}_4$  to fcc and hcp Ru, further indicating the charge transfer behavior is not affected by altering the

crystalline phases of Ru cocatalysts. Thus the phase-dependent photocatalytic performance in reduction of  $\text{CO}_2$  mainly attributes to the different interactions between  $\text{CO}_2$  and the surface of Ru cocatalysts.

#### 4. Conclusions

In summary, crystalline phase engineering was performed on the Ru cocatalysts for enhanced photocatalytic performance in the conversion of  $\text{CO}_2$  into fuels. Ru cocatalysts in fcc and hcp phases acting as cocatalysts were deposited on the flat surface of  $\text{C}_3\text{N}_4$  nanosheets, respectively. It turns out that the hcp Ru cocatalysts realize the superior average CO and  $\text{CH}_4$  production rate but lower  $\text{H}_2$  production rate in comparison with fcc Ru. As a result, the selectivity of  $\text{C}_3\text{N}_4$ -fcc Ru in  $\text{CO}_2$  reduction is significantly higher than that of  $\text{C}_3\text{N}_4$ -hcp Ru (71.8% versus 46.5%). It was found that the phase-dependent photocatalytic performance is not resulted from the varied electron transfer abilities from  $\text{C}_3\text{N}_4$  to Ru, but the different surface reaction kinetics on the surface of fcc and hcp Ru cocatalysts. Theoretical simulations indicate that the  $\text{CO}_2$  adsorption energy on the dominated (10-11) face of hcp Ru is higher than that on the dominated (111) face of fcc Ru. Thus the stronger interaction between  $\text{CO}_2$  molecules and surface of hcp Ru cocatalysts contributes to the enhanced photocatalytic activity and selectivity of  $\text{C}_3\text{N}_4$ -hcp Ru. This work presents an important advancement towards the design of cocatalysts for enhanced photocatalytic performance through crystalline phase engineering.

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#### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.apcatb.2018.02.046>.

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